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Polymer Distributed Bragg Reflectors for Vapor Sensing

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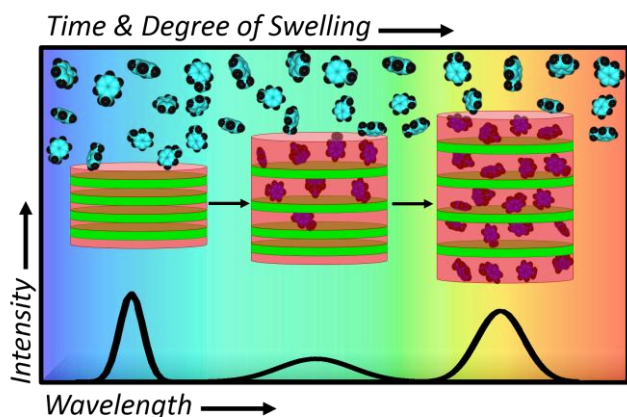
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ABSTRACT: We establish a new strategy to achieve fast and responsive hybrid distributed Bragg reflectors for environmental vapor sensing. We fabricated easily processable zinc oxide-polystyrene nanocomposites to grow high quality multilayers with large gas permeability and dielectric contrast, which enable fast and sensitive detection of vapor analytes. Multilayers fabricated by simple spin-coating of the nanocomposite and cellulose acetate show a 10-fold enhancement of the optical response to toluene exposure compared to reference distributed Bragg reflectors built with bare polystyrene.

KEYWORDS: 1D photonic crystals, polymer DBR, ZnO nanoparticles, optical nanocomposites, photonic crystal sensors

TOC GRAPHIC



INTRODUCTION

Distributed Bragg reflectors (DBRs) are widely used in optoelectronics as antireflective coatings for photovoltaic devices, resonators for lasing cavities, light manipulation in light emitting diodes and active material in biological and chemical sensors.¹⁻⁷ Thanks to their good mechanical properties, low cost and ease of fabrication (e.g. by self-assembly,⁸ coextrusion,⁹ or spin-coating³), polymer-based DBR are attractive for the production of low-cost, disposable molecule sensors. Molecular detection has already been demonstrated in photonic crystals, where the modification of the optical response is caused by analytes permeation within the structure and, at times, bonding with chemically active additives.^{10,11} The high sensitivity of photonic crystal sensors is related to their photonic band structure, which describes light interaction in a medium with periodic modulation of the refractive index. Forbidden energy regions in the photonic crystal band structure inhibit propagation of isoenergetic light frequencies and generate photonic band gaps whose characteristics strongly depend on the refractive index contrast and its spatial modulation.^{12,13} Penetration of analyte molecules within the crystal modifies the effective refractive index, thus affecting spectral position, intensity and width of the photonic band gaps, which can be detected optically.

Despite the widespread use of polymers as building blocks in semi-industrial fabrication processes,^{9,14} so far their use for fabrication of DBR vapor sensors has been hindered by two main issues: first, while highly permeable porous lattices such as opals,^{10,15-21} colloidal crystals,^{11,22,23} porous inorganic^{24,25-26,27} and hybrid^{28,29} DBR are highly sensitive to both liquid and gas analytes, permeability to gas and vapors of polymer DBR is limited by low porosity of amorphous polymers. This may be overcome by sophisticated optical configurations for the detection of vapors.³⁰⁻³³ However, just a few vapor sensitive polymer DBR have been reported so far: in 2003, Convertino *et al.* showed acetone vapor responsive DBRs made of alternated layers of bare and gold doped poly(tetrafluoroethylene) fabricated by chemical vapor deposition.³⁴ In 2006 Mönch *et al.* described a multilayer grown by spin-coating of monomeric precursors of poly(methylmethacrylate) and polystyrene and *in situ* polymerization by ultraviolet curing, achieving good sensitivity to acetone and toluene vapors.³⁵

The second issue is related to the low dielectric contrast typical of mutually processable polymers. Even though polymer refractive indices in the visible range vary from $n \approx 1.3$ for fluorinated polymers to $n \approx 2.0$ for conjugated systems, co-processing is restricted by their chemical nature. For instance, co-extrusion requires high chemical compatibility which is hardly achievable with fluorinated polymers.⁹ On the other hand, spin-coating requires complete orthogonality between polymers and solvents of alternating layers to prevent dissolution of the underlying films.² Moreover, high absorbance hinders the use of conjugated systems in the visible spectral range and crystalline and semi-crystalline polymers scatter light substantially, which is detrimental for optical applications.

In this work we fabricated flexible, free-standing polymer DBRs made of cellulose acetate ($n \approx 1.46$) and polystyrene ($n \approx 1.57$) loaded with high refractive index nanoparticles (Figure 1), and demonstrate their high sensitivity to toluene vapors. In this design, the filler performs a dual function: i. To introduce free volume in the polystyrene matrix thus increasing its vapor permeability.^{36,37} ii. To increase the matrix effective refractive index (given by the volume-weighted average of the dielectric components) and overall the DBR refractive index contrast.^{34,38} Meanwhile, thanks to the negligible scattering contribution, the nanoparticle filler preserves high transparency of the loaded films.

As a proof of concept, high refractive index ZnO nanoparticles (NPs, $n \approx 1.98$) were synthesized by ultrasound-assisted solvothermal synthesis and properly functionalized to be dispersed in polystyrene. The nanoparticles were then incorporated in a well-studied polymer DBR structure comprising of spun-cast alternating layers of polystyrene and cellulose acetate.^{2,39} Incorporation of optical nanocomposites turns out to be an effective strategy to overcome the major limitations of conventional polymer DBRs: the possibility to increase the dielectric contrast of well-known spin-coatable polymer pairs^{2,39,40} and their permeability at the same time opens up new perspectives for polymer photonic crystal sensing.

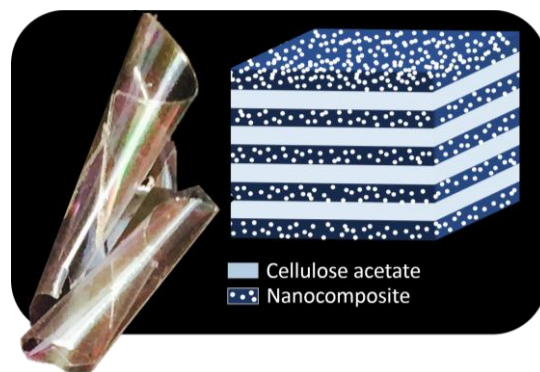


Figure 1: Schematic and photograph of a flexible DBR fabricated with cellulose acetate and polystyrene-ZnO nanoparticles nanocomposite.

RESULTS AND DISCUSSION

The optical nanocomposite used for DBR fabrication was prepared by dynamic spin-coating of ZnO nanoparticles dispersion in polystyrene-toluene solutions. The nanoparticles were synthesized by a solvothermal route (see experimental section and supporting

information for details). Their average size is 14 nm, with standard deviation of 3.6 nm, as determined by scanning electron microscopy (SEM, see inset of Figure 2b and Supporting Figure S1a). X-ray powder diffraction pattern indicated 13 nm crystals in Wurtzite phase⁴¹ as derived from Debye-Scherrer equation data treatment⁴² and diffraction peak indexing (Supporting Figure S1b).

To achieve good dispersion within the non-polar matrix, the particles were grafted with a monolayer of dimethyl-(methoxy) octadecylsilane (DMMOS). Figures 2a and 2b compare the SEM micrographs collected for thin polystyrene film loaded with 2.5% v/v_{PS} of bare and grafted ZnO nanoparticles. While several aggregates of few micrometers size are present in the film prepared with bare nanoparticles, the surface of the film containing the grafted particles appears smooth, without signs of aggregation. Negligible light scattering is thus expected from this nanocomposite.

The complex refractive index ($\tilde{n}=n+ik$) of the thin film nanocomposite derived by spectroscopic ellipsometry confirms the negligible light scattering (Figure 2c). Here, the bare polystyrene extinction coefficient value is zero in the entire spectral range and its refractive index dispersion agrees with literature data ($n=1.581$ at 800 nm).² For the nanocomposite, a weak peak due to ZnO absorption is visible at 380 nm in the dispersion of k . The real part of its refractive index is slightly higher (by less than 0.5%) than the one of polystyrene at long wavelengths ($n=1.587$ at 800 nm), while it is nearly identical in the UV. This confirms that a relatively low concentrations of high refractive index nanoparticles (2.5% v/v_{PS}, 14% w/w_{PS}) does not affect the film transparency. Only at larger nanoparticle concentrations (e.g. 5 and 10% v/v_{PS}) light scattering becomes significant and prevent the formation of a photonic band gap.

The increase of permeability induced by the loading in the polystyrene matrices was evaluated by pressurized helium permeation tests on free-standing nanocomposite and bare polystyrene thin films (see experimental section). The molar gas flow passing through the sample is related to the permeability (PE) by:

$$PE = \frac{Ql}{\Delta P D} \quad (1)$$

where Q is the He molar flow, ΔP is the He gas pressure, l and D are the film thickness and the permeation area, respectively (see experimental section for details). Figure 2c shows that the permeability of bare polystyrene is very low, and does not depend on applied pressure. Remarkably, the permeability of the nanocomposite undergoes a 3.5 fold increase and shows strong dependence on applied pressure. This behavior can be attributed to free-volume introduced at the nanoparticle-polystyrene interface, which facilitates gas permeation.

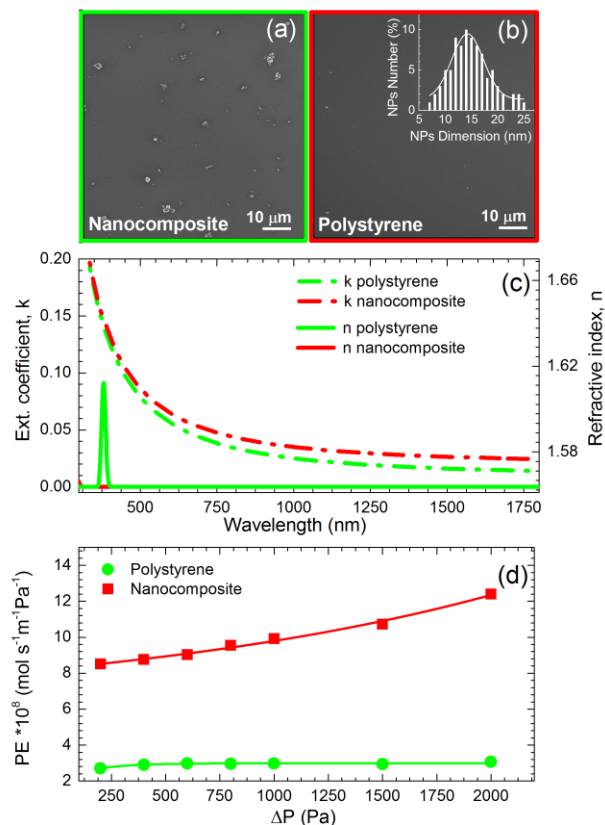


Figure 2: SEM micrographs of nanocomposite thin films loaded with: a) bare ZnO nanoparticles and b) nanoparticles grafted with DMMOS. Inset shows the nanoparticles size distribution retrieved from SEM data. c) Refractive index (dash line) and extinction coefficient (full line) dispersions for bare polystyrene (green) and nanocomposite (red). d) Permeability to He gas of polystyrene (green dots) and nanocomposite (red squares) thin films as a function of applied pressure. ZnO nanoparticles concentration in the nanocomposite is 2.5% v/v_{ps}.

DBR multilayers were fabricated by alternated spin-coating deposition of cellulose acetate and nanocomposite with ZnO nanoparticles concentration of 2.5% v/v_{ps}. Figure 3 (red line) shows the reflectance spectrum of a 15 periods DBR with maximum reflectance peaks at 1505 nm (λ_{PBG}), 758 nm ($\lambda_{\text{PBG}}/2$), 506 nm ($\lambda_{\text{PBG}}/3$), 386 nm ($\lambda_{\text{PBG}}/4$) and 317 nm ($\lambda_{\text{PBG}}/5$). These peak wavelengths correspond to the first order photonic band gap and higher order replicas of the DBR. The baseline of the reflectance spectrum shows an interference pattern typical of high optical quality structures, which arises from the interference of waves partially reflected by the front and back surfaces of the DBR. The DBR optical response was modeled by a home-made software code based on the transfer matrix formalism, using measured refractive indices (Figure 2b and ref.²) and layer thicknesses (Supporting Figure S2) as input. The good agreement between experimental and simulated spectra in Figure 3 confirms the excellent optical quality of the multilayer. Minor discrepancies appear in the UV spectral range, where thickness variations, refractive index precision, and spectrometer sensitivity become critical.

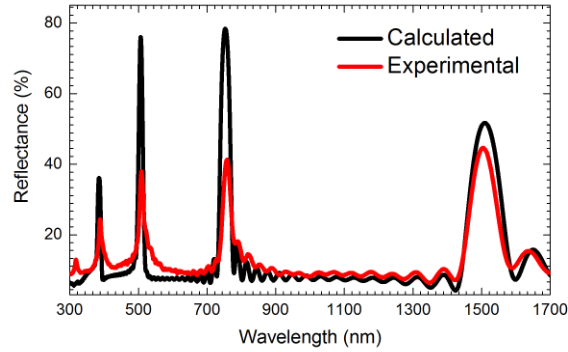


Figure 3: Comparison between experimental (black) and calculated (red) reflectance spectra of a 15 period nanocomposite – cellulose acetate DBR.

Figure 4 shows the nanocomposite DBR transmittance spectra recorded as a function of incidence angle for *p* and *s* light polarization. For both polarizations, the five Bragg peaks shift toward smaller wavelengths increasing the angle of incidence. While in *s* polarization the width of Bragg peaks is nearly independent of the angle of incidence, in *p* polarization both width and intensity of the peaks decrease toward the Brewster angle. This is consistent with the assignment of photonic band gaps and fully agrees with theoretical predictions (see calculated spectra in Supporting Figure S3).²

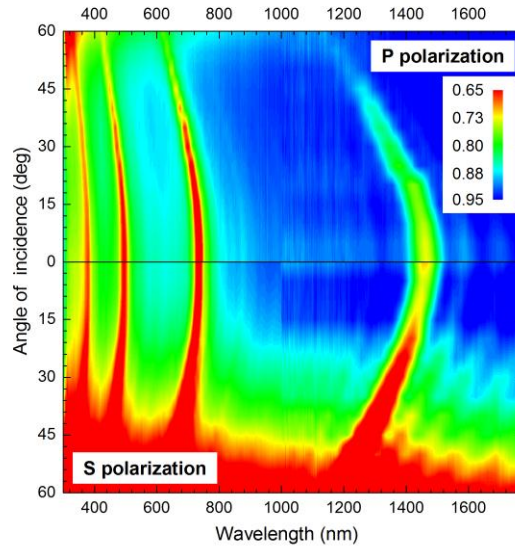


Figure 4: Transmittance spectra contour plot as a function of light incidence angle for *p* (top panel) and *s* (bottom panel) light polarization. The color scale represents transmitted intensity.

To evaluate the sensing performance of the nanocomposite DBR, the sample was exposed to toluene vapor in a closed cell and its optical response was compared to that of a control sample fabricated using bare polystyrene with similar film thickness (Figs. 5). In

the case of the nanocomposite DBR (Figure 5a), the transmittance spectra show a remarkable spectral shift of the photonic band gap of ~50 nm after 15 minutes of exposure (Figure 5b), and a ~20% reduction of the band gap intensity in differential transmittance (Figure 5c). Conversely, no significant changes are observed in either transmittance or differential transmittance spectra of standard polystyrene DBR over time (Figure 6b' and 6c')

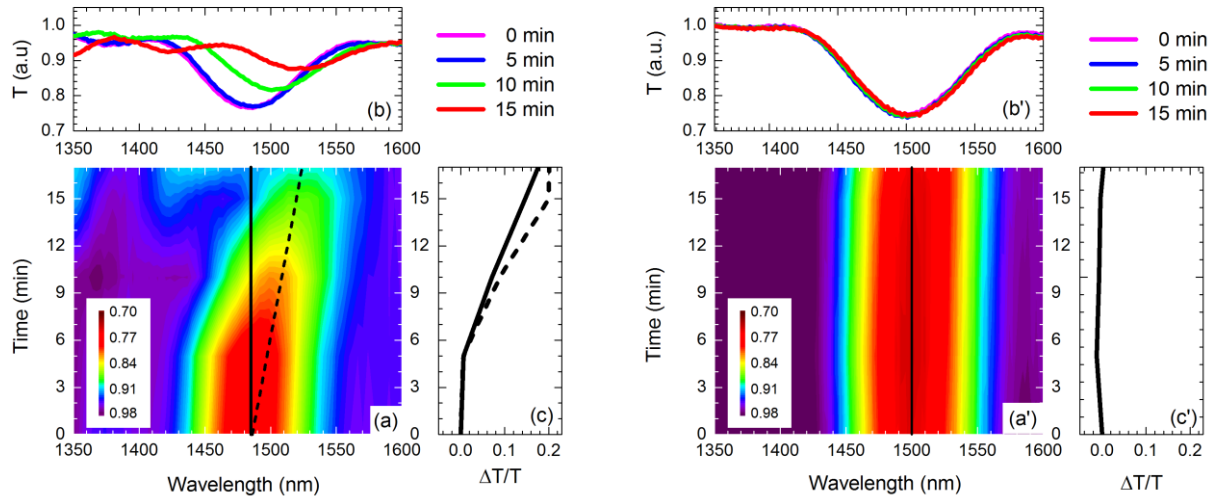


Figure 5: Optical response to toluene vapor exposure of DBR sensors. Nanocomposite DBR on the left and control DBR on the right. a), a') Contour plot of the transmittance spectra collected every 5 minutes; b), b') Corresponding transmittance spectra; c), c') Time dependent differential transmittance at initial photonic band gap wavelength (solid line in panel a and a') and minimum of transmission (dashed line in panel a).

The optical response of the DBR vapor sensor at longer exposure times is even richer (Figure 6). After the first 15 minutes of exposure to toluene vapor, the transmittance peaks of the nanocomposite DBR undergo non-monotonic spectral and intensity changes (Figure 6a), while those of the reference polystyrene multilayer show minimal variations in intensity and a only a slight red-shift (Supporting Figure S5). The contour plot in Figure 6a clearly shows transmittance peaks related to the photonic band gap of the DBR (marked by the * symbol), as well as fringes related to the interference of light reflected by the front and back faces of the multilayer structure. The photonic band gap peak red-shifts for the first 15 minutes (compare to Figure 5a), disappears thereafter, and reappears after ~80 minutes at about 1850 nm (marked by the *' symbol). After two hours, the spectrum still evolves slowly (data non shown here). When the same measurements are repeated upon diacetone alcohol (cellulose acetate solvent) vapor exposure, no effects are observed in the optical spectra of both polystyrene and nanocomposite multilayers. Toluene and diacetone alcohol exposure data are compared in Supporting Figure S5.

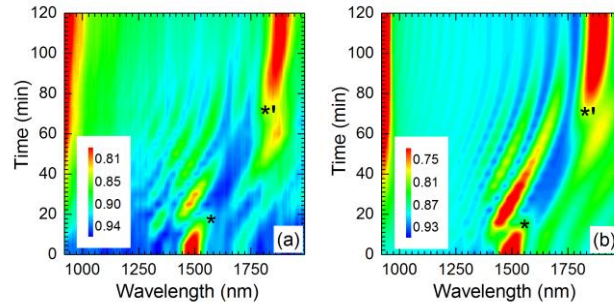


Figure 6: Contour plots of the transmittance spectra collected during the exposure for a) nanocomposite DBR and b) control sample. Intensity is represented as a color scale.

The spectral shift of the photonic band gap and its intensity modulation are due to the swelling of the permeable nanocomposite layers induced by toluene molecules that penetrate into the DBR. Indeed, the complex optical response can be reproduced exceptionally well by a fairly simple phenomenological model accounting for the increase of thickness of the nanocomposite layers (D_{NC}) upon permeation of solvent vapors from the surface in contact with the toluene rich environment. Let us consider a progressive swelling of the nanocomposite layers over the time (t), while the cellulose acetate layer thicknesses (D_{CA}) remain constant:

$$D_{I,CA}(t) = D_{j,CA}(0) \quad (2)$$

$$D_{j,NC}(t) = D_{j,NC}(0) + \int_{t_0}^t \frac{\partial D_{j,NC}(t)}{\partial t} dt \quad (3)$$

Under these assumptions, the thickness of each nanocomposite layer j increases over time according to its swelling velocity ($v_j = \partial D_{j,NC}(t)/\partial t$).

The system of equation 2 and 3 was solved numerically for discrete time intervals, using the initial layer thicknesses measured by interference microscopy (Supporting Figure S2) as boundary conditions and arbitrarily setting time scale to fit the experimental data.

Note that this model makes two assumptions: 1. The thickness of cellulose acetate layers does not vary with time, and permeation across cellulose acetate occurs instantaneously (one could introduce a time delay due to permeation across cellulose acetate layers, but this would not affect the swelling velocities v_j and their physical meaning). 2) The variation of the refractive index of the layers upon permeation of solvent vapors is accounted by the variation of the effective optical thickness ($\partial D_{j,NC}(t)/\partial t$), which is defined as the product $D_j n_j$.

The time evolution of each nanocomposite layer thicknesses and swelling velocities derived from the model are reported in Figure 7. The thickness of the first layer increases quickly upon toluene exposure; its swelling velocity is very fast initially due to permeation of toluene, and then decreases (Figure 7b) until the thickness becomes ~25% larger than the original value (Figure 7a). Toluene permeates progressively into the inner layers of the structures, which then undergo similar swelling process. After about one hour, the outermost half of the DBR is fully swollen while the innermost half is still swelling. Steady-state is reached only after several hours of exposure.

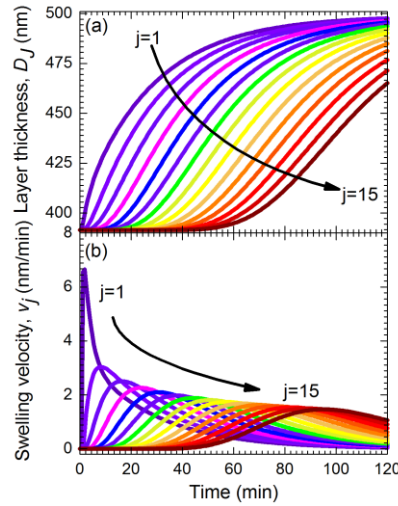


Figure 7: Evolution of the modeling parameters of each nanocomposite layer in the DBR over exposure time: a) layer thickness and b) swelling velocity.

The calculated layer thicknesses were used to model the time evolution of the spectral response of the DBR. The good agreement between experimental (Figure 6a) and calculated (Figure 6b) spectra indicates that the model is able to capture the macroscopic physical changes undergone by the sample, and that progressive swelling of the DBR layers nearest to the surface (close to the toluene rich environment) is indeed responsible for the spectral shift of the DBR peaks. Permeation of toluene vapors across the DBR layers modifies their optical thickness, resulting in a rather complicated but predictable evolution of the transmittance spectra. Within the first 15 minutes of exposure, the photonic band gap peak red-shifts due to the thickness increase of the top DBR layers (Figure 6 marked by the symbol * and Fig 7a). Upon toluene penetration in the underlying layers, the diffraction patterns of differently swollen films interfere destructively, and the intensity of the photonic band gap peak is progressively reduced until it cancels after ~20 minutes of exposure (Figure 6 between the symbols * and *' and Figure 7a). Only after ~80 minutes of vapor exposure, when a significant part of DBR layers reaches a stationary swollen condition, the restored optical periodicity results in the appearance of a new, clearly defined band gap (Figure 6 marked by the *' symbol and Figure 7a). We suggest that free volume at the ZnO-polystyrene interface may behave as solvent condensation point, initiating the swelling process, which then propagates to the whole nanocomposite layer.

As mentioned earlier, similar effects have been observed in self-assembled electrolytes block co-polymer multilayers exposed to solution containing different ions⁴³ as well as in highly porous opal structures but, to the best of our knowledge, this is the fastest responsivity to vapors reported for entirely solution-processed DBR prepared by spin-casting. Furthermore, within this scheme analytes selectivity could also be achieved by grafting a silane with selective functional group onto the ZnO nanoparticles surface.

CONCLUSIONS

We reported new polymer DBR sensors with high optical responsivity to toluene vapor exposure. Vapor sensitivity was enhanced by increasing the overall DBR permeability with a new polystyrene-ZnO nanocomposite which show permeability 3.5 time higher than bare polystyrene. DBRs displaying up to the fifth diffraction order were fabricated by simple spin-coating of alternated layers of

cellulose acetate and the new material, which have slightly higher refractive index than bare polystyrene. The comparison between nanocomposite and polystyrene DBR sensing performance shows that the filler induces a ~ 10 time faster optical response to toluene vapors exposure. The better response is due to presence of free-volume induced by the loaded nanoparticles, which allows faster swelling of the polystyrene. The complex dynamic of vapor permeation and swelling of each DBR layer was fully understood in terms of a simple analytical model, which allows predicting the overall DBR optical response up to very long time of exposure to the target vapor. This provides a very robust platform for the design of disposable vapor and gas sensors that are fast, sensitive and selective, for applications in environmental and safety related monitoring such as the detection of hydrocarbons on ocean surface as well as toxic and pollutant gas and vapors in industrial areas.

METHODS

Zinc oxide nanoparticles synthesis and functionalization: ZnO nanoparticles were synthesized via a solvothermal route starting from zinc acetate dihydrate and potassium hydroxide. In a typical process, 0.07 mol of zinc acetate are dissolved in methanol (MeOH) and heated at 63 °C under sonication. Then, 0.14 mol of KOH are dissolved in the same solvent and slowly added to the first solution. After 3 hours of reaction, the particles are purified by 5 series of decantation and rinsing with MeOH, and eventually desiccated.⁴⁴ In order to prevent aggregation of ZnO nanoparticles, which is responsible for light scattering in the non-polar polystyrene, DMMOS was grafted onto nanopartilces surface. To achieve the grafting, 10g of nanoparticle were dispersed in 30 ml of MeOH and sonicated. Then, 70 ml of dichloromethane containing 2.5 g of DMMOS were added to the dispersion. The overall dispersion dried and desiccated for 2 h at 40 °C in a vacuum. The graft reaction between the surface and the adsorbed DMMOS was run at 135°C under nitrogen flux for 2 h. nanoparticle growth and grafting reaction mechanism are reported in Supporting Information.

Nanocomposite and DBR preparation: The grafted ZnO nanoparticles were dispersed into toluene solutions of polystyrene ($M_w = 200000$) and stirred for more than 36 h. The new colloidal dispersion and bare polymer solutions were used to grow both thin films and DBR by spin-coating.^{2,3} Multilayers were fabricated alternating films of nanocomposite or polystyrene and cellulose acetate ($M_w = 61\ 000$). The polymer concentrations ranged from 2,8 to 3,5 % (v/v) and the rotation speed during the deposition was kept at about 6000 revolutions per minute. Thin film thicknesses was measured on mono and bilayers made by nanocomposite and cellulose acetate by light interferometry using a GBS smart WLI microscope with a 20x interference objective.

Materials and DBR characterization: ZnO nanoparticles and the new nanocomposite thin films were analyzed by SEM using a field emission scanning electron microscope Jeol JSM6700F. The nanoparticles crystal size and structure was identified by XRD. Step scan powder diffraction patterns were collected with a D8 Advance Bruker.

The polymers complex refractive index was determined by spectroscopic ellipsometry with a VASE instrument by J.A. Woollam Co. Inc. and analyzed using the commercial software WVASE32®. Incidence angles ranging from 55° to 75° have been used.

Thin films permeability were measured with a home-made system schematized in Supporting Figure S6 In this analysis, Helium is pressurized and permeated trough free standing polymer films. The gas flow is measured using a flow meter and related to the sample permeability by Equation 1.

DBR reflectance and transmittance spectra were collected with setups based on optical fiber using an Avantes AvaSpec-2048 spectrometer (200–1150 nm, resolution 1.4 nm) and an Arcoptics FT-interferometer (1000-2600 nm, resolution 8cm⁻¹). The light source was a combined deuterium–halogen Micropak DH2000BAL.

SUPPORTING INFORMATION

ZnO nanopartilces SEM characterization and X-ray diffraction pattern. Cellulose acetate and nanocomposite film thicknesses profile as measured from refraction microscopy. Calculated angle dispersion transmission spectra of nanocomposite DBR. Comparison between DBRs optical response to diacetone alcohol and toluene vapors. Thin film permeability measurements set-up. ZnO nanoparticles synthesis and grafting reaction mechanisms.

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